実験報告書様式(一般利用課題·成果公開利用)

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課題番号 Project No.	装置責任者 Name of responsible person
2008A0010	Yasuhiro Miyake
実験課題名 Title of experiment	装置名 Name of Instrument/(BL No.)
Lithium diffusion in lithium-transition-metal-oxides	D1
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試料、実験方法、利用の結果得られた主なデータ、考察、結論等を、記述して下さい。(適宜、図表添付のこと) Please report your samples, experimental method and results, discussion and conclusions. Please add figures and tables for better explanation.

1. 試料 Name of sample(s) and chemical formula, or compositions including physical form.

Lithium nickel dioxides, $Li_{1-x}Ni_{1+x}O_2$ with x~0 and x~0.07.

A powder sample was pressed in a disc with 30 mm diameter and 2 mm thickness, and then the disc was packed in an Au-sealed cell.

2. 実験方法及び結果(実験がうまくいかなかった場合、その理由を記述してください。)

Experimental method and results. If you failed to conduct experiment as planned, please describe reasons.

In order to confirm the unique power of mSR for detecting the diffusion coefficient of Li+ ions (D_{Li}) in solids, we have measured ZF-, wLF-, and wTF-mSR spectra for lithium nickel dioxides, which were heavily investigated as a cathode material for the next-generation Li-ion batteries, in the temperature range between 60 and 450 K. In the rhombohedral LiNiO₂ lattice with space group R3-m, the NiO₂ plane and the Li layers form alternating stacks along the c_{H} -axis in the hexagonal setting (see Fig. 1). In the NiO₂ planes, Ni ions form a two-dimensional triangular lattice (2DTL) by a network of edge-sharing NiO₆ octahedra. In contrast to LiCoO₂, a stoichiometric LiNiO₂ has been never prepared so far. That is, the excess Ni is usually present in the Li layer of the LiNiO₂ samples due to the similarity in ionic radii between Li⁺ and Ni³⁺. The ionic distribution of the Ni-excess LiNiO₂ is thus given by (Li⁺_{1-x}Ni²⁺_x)_{3b}[Ni²⁺_xNi³⁺_{1-x}]_{3a}O₂, where 3b and 3a are the Li and Ni site in the regular LiNiO₂ lattice. Besides of an interesting change in low-*T* magnetism of Li_{1-x}Ni_{1+x}O₂ with *x*, the (Ni²⁺)_{3b} in series are naturally expected to affect D_{Li} at high *T*, because of planer hindrance of (Ni²⁺)_{3b} in the Li layer, in which Li⁺ ions move relatively

2. 実験方法及び結果(つづき) Experimental method and results (continued)

easily. Nevertheless, there is, to authors' knowledge, less systematic work on the relationship between D_{Li} and x, mainly due to lack of a proper tool for detecting D_{Li} . The μ SR experiment on $\text{Li}_{1-x}\text{Ni}_{1+x}\text{O}_2$ with $x\sim0.03$ and ~0.15 , therefore, provides crucial information on the effect on D_{Li} , resulting in clear insight how to improve cathode materials.

In fact, both ZF- and LF-spectra for the *x*~0.03 sample were well fitted by a dynamic Kubo-Toyabe function, which is expected for a random nuclear magnetic field due to mainly ⁷Li. By using a global fit technique for ZF-, LF-, and wTF-spectra, the field distribution width (Δ) and field fluctuation rate (v) were estimated at each temperature (Fig. 2). As temperature increases from 70 K, v looks almost *T*-independent (~0.2 x 10⁶ s⁻¹) up to 250 K, and then increases rapidly with increasing *T* until ~370 K, and finally decreases with *T*. That is, the v(*T*) curve exhibits a peak around 370 K. On the contrary, Δ decreases linearly with *T* below 200 K, and then Δ seems to level off to ~0.28 x 10⁶ s⁻¹ up to ~300 K, and finally decreases with *T* until 450 K.

Assuming that v-v(T<250 K) corresponds to the jump rate of the Li ions between the neighboring sites, D_{Li} is estimated as ~8 x 10⁻¹² cm²/s at 370 K. This value is quite large compared with D_{Li} obtained by the ⁷Li-NMR experiment (6 x 10⁻¹⁵ cm²/s at 500 K). This is probably due to the effect of the Ni spins on the ⁷Li nuclear spin-lattice relaxation rate of NMR, particularly of the Ni²⁺ ions in the Li plane.

In order to confirm that v really corresponds to the jump rate of the Li ions, we also planed to measure the sample with $x\sim0.15$. This is because the Ni²⁺ ions in the Li plane naturally reduce D_{Li} . However, because of the unexpected low power of the proton/muon beam in J-PARC, the counting rate was almost 1/10 to that of ISIS, where it takes 2 hours per one *T* point. Therefore, although we used 13 shifts in total, we measured only three *T* points for the $x\sim0.15$ sample. The preliminary result looks to be consistent with our expectation, but we need more data to know the *T* dependences of both v and Δ for the $x\sim0.15$ sample in detail. Such experiment will be performed in the



Fig.1 The crystal structure of LiNiO₂.



Fig. 2 Temperature dependences of (top) field fluctuation rate n and (bottom) field distribution width for $\text{Li}_{1-x}\text{Ni}_{1+x}\text{O}_2$ with *x*~0.03 and ~0.15. The data were obtained by global-fitting the ZF-, LF-, and wTF-spectra.